# Investigation of the Influence of Al<sub>2</sub>O<sub>3</sub> Particles on the **Microhardness and Tensile Strength of PA6 Composites**

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**Abstract:** Polyamide is a high-performance synthetic plastic known for its strength, durability, flexibility, chemical resistance, and low cost, making it widely used in engineering, automotive, and electrical. However, the surface and mechanical properties can be further enhanced to meet the growing demands of advanced engineering applications. This study aims to investigate the influence of  $Al_2O_3$  particles on the hardness of polyamide 6 (PA6). The  $Al_2O_3$  was mixed with PA6 at weight percentages (wt.%) of 0.3% and 1.5% then were fabricated into composite plates using compression molding and subsequently. As a result, the composites achieved higher microhardness and tensile strength compared to the matrix with increases of 13.3% and 7.3% achieved by incorporating 0.3 wt.% of reinforcement, respectively. This result suggests that  $A_2O_3$  has the potential to improve the surface properties and mechanical strength of the matrix material.

**Keywords:** Al2O3, compression molding, microhardness, polyamide 6, tensile strength.

# **1. INTRODUCTION**

Polyamide 6 (PA6), commonly known as nylon 6, is a widely used thermoplastic polymer known for its excellent mechanical properties, thermal stability, and resistance to chemicals and rust. PA6 is a semicrystalline thermoplastic polymer widely utilized in engineering applications [1, 2]. It is highly regarded for its excellent processability, strong tensile properties, and outstanding resistance to abrasion and chemical agents [3]. These properties make it a material of choice for various industrial applications, including automotive components, electrical and electronic devices, and consumer goods [4]. However, the inherent mechanical properties of PA6 can be further enhanced to meet the growing demands for highperformance materials in advanced engineering applications [5].

One effective strategy to improve the mechanical properties of polymers is the incorporation of particulate reinforcement. PA6 properties can be improved by incorporating reinforcement. Bragraglia et al. utilized boron nitride (BN) as a reinforcement. The addition of 45 wt.% BN to PA6 results in a 153% increase in elastic modulus. However, when compared to pure PA6, both the strain at rupture and ultimate tensile strength (UTS) decrease by as much as 50.7% [6]. The most recent research aimed at enhancing the PA6 matrix with microparticles was conducted by Tanoto and Huang, who utilized AZ61 particles as reinforcement within the PA6 matrix. The maximum values achieved for tensile strength and hardness

were 58 MPa and 21.1 HV, respectively. These peak values were attained by incorporating 3 wt.% and 5 wt.% of AZ61 into the matrix [7].

Among the various particles studied, alumina  $(AI_2O_3)$  stands out due to its exceptional hardness, high thermal conductivity, and stability.  $Al_2O_3$  particles, being a type of metal oxide, offer excellent properties that can greatly enhance and improve various characteristics of polymer composites, making them a promising option for polymer reinforcement [8]. The tribological analysis, studied by Ain *et al*. shows that increasing  $Al<sub>2</sub>O<sub>3</sub>$  content reduces the friction coefficient by enhancing load distribution across the interface of the Polydimethylsiloxane (PDMS) matrix [9]. Coban *et al*. report that the high modulus of  $Al_2O_3$  and its strong interaction with the Polyphenylene sulfide (PPS) matrix enhance dynamic mechanical properties [10]. Rawat *et al*. reported that alumina micro powder effectively increased the hardness of glass fiber by as much as 10% [11]. However, no one paper discussed Al2O3 as a reinforcement for PA6.

The literature review indicates that the incorporation of polymer and  $Al_2O_3$  presents significant potential for mixing as a polymer matrix composite (PMC). The  $Al_2O_3/PAG$  composite is promising as a lightweight material with excellent surface hardness and mechanical properties. It is important to note that this study specifically focuses on the impact of  $Al_2O_3$ as a reinforcement material to enhance the properties of the PA6 matrix.

# **2. EXPERIMENTAL**

### **2.1. Material**

In this study, the same PA6 powder from the previous study was utilized as the matrix material [7].

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For reinforcement,  $Al_2O_3$  micro powder with an average particle size of 7.5  $\mu$ m (D<sub>10</sub>= 0.24  $\mu$ m, and  $D_{90}$ = 17.5 µm) was employed. Al<sub>2</sub>O<sub>3</sub> was incorporated into the matrix at various weight percentages, specifically 0.3 wt.%, and 1.5 wt.%. These two values were selected based on the results of preliminary experiments. With a composition difference reaching fivefold, the effect of Al2O3 on the material's performance was clearly observed.

# **2.2. Composite Preparation**

The PA6 and  $Al_2O_3$  powders were mixed in varying weight percentages of 0.3 wt.% and 1.5 wt.%  $Al_2O_3$ using Retsch PM100 planetary ball milling apparatus running at 300 rpm. The ball-to-powder ratio (BPR) was maintained at 1:1 to ensure uniform mixing of the powders. After ball milling, the powder mixtures were processed into composite plates through compression molding [9, 10].

A square mold with dimensions of 11 cm × 11 cm and a thickness of 1 mm was used to form the plates. Approximately 13 grams of the powder mixture was used for each composite. Various compression molding parameters were tested, including temperature, pressure, and compression time. The temperatures tested were 210°C, 215°C, and 220°C (the melting temperature of PA6), with a constant pressure of 7 MPa. Compression times of 5 minutes and 7 minutes were evaluated to determine the optimal conditions.

Before full compression, a preheating stage was implemented to minimize the presence of moisture in the powder mixture and improve material flow [14]. Here the temperature was gradually increased from 190°C to the target temperature of 215°C. This preheating process ensured proper fusion of the materials before full compression. After testing several samples, the optimal parameters were identified as 215°C for 7 minutes of compression.

Following the main compression molding stage, a cooling down stage was employed where the plates were allowed to cool down from 215°C to 180°C under the same pressure of 7 MPa. The pressure was later released and the mold was taken out after allowing it to cool down to room temperature.

# **2.3. Characterization**

The morphology of PA6 and  $Al_2O_3$  powder was investigated using a field emission scanning electron microscope (FE-SEM) model JSM-7900 from Jeol Ltd., Japan. This same instrument was also utilized to examine the elemental mapping of the composite powder. Surface properties were observed using a microhardness tester, Wilson HV1102, with an applied pressure of HV 0.1 kgf for a 10-second indentation period. The tensile test was conducted in accordance with ASTM D638 standards using Type V specimens, with a specimen thickness of 1 mm. The tests were performed on an MTS Insight Electromechanical Standard Length (SL) machine, equipped with a 10 kN load capacity.

# **3. RESULT AND DISCUSSION**

# **3.1. Material Characterization**

As seen in Figure **1**, the morphology of PA6 particles is characterized by irregular shapes with sharp, distinct edges rather than spherical forms [15]. Similarly,  $Al_2O_3$  particles also exhibit irregular, nonuniform structures.

The morphology and elemental distribution of the composite powder after mixing were examined using SEM imaging, as presented in Figure **2**. The SEM image in Figure **2a** illustrates the composite powder containing 0.3 wt.% reinforcement at a magnification of 750x. PA6 was identified by the presence of oxygen (O) (Figure **2c**) and nitrogen (N) (Figure **2d**) elements, while  $Al_2O_3$  was primarily composed of aluminum (Al) (Figure **2b**) and oxygen (O). As seen in Figure **2b**, the



**Figure 1:** SEM images of (a) PA6 and (b) Al<sub>2</sub>O<sub>3</sub> as received.



**Figure 2:** SEM images of 0.3 wt.% of Al2O3/PA6 (**a**) 750 magnifications. EDS mapping of (**b**) aluminum distribution, (**c**) oxygen distribution, and (**d**) nitrogen distribution.

Al<sub>2</sub>O<sub>3</sub> particles are well integrated into the Al<sub>2</sub>O<sub>3</sub>/PA6 powder blend, with aluminum being uniformly dispersed across the PA6 particles, indicating successful reinforcement mixing within the polymer matrix.

The SEM image in Figure **3a** illustrates the composite powder containing 1.5 wt.%. Figure **3b** shows the aluminum element, meanwhile, oxygen and nitrogen elements are shown in Figure **3c** and **3d** respectively. The SEM and EDS images demonstrate that at higher concentrations of  $Al_2O_3$ , the particles tend to agglomerate. The biggest cluster of  $Al_2O_3$ particles is highlighted by the blue arrow pointing to regions of concentrated aluminum in the SEM images. The agglomerated  $Al_2O_3$  is further confirmed by the EDS analysis, where the green and red colors, representing the aluminum (Al) and oxygen (O) elements, exhibit a higher intensity in those specific areas. This suggests that the  $Al_2O_3$  particles cluster in certain regions, leading to localized accumulations of the reinforcement within the PA6 matrix. A higher concentration of powder reinforcement often leads to increased agglomeration. This phenomenon occurs because as more particles are added to the polymer matrix, the likelihood of particle-to-particle contact rises, which can result in clustering and the formation of agglomerates [16].

#### **3.2. Surface Properties**

The microhardness values for pure PA6 and Al2O3/PA6 composites at various weight percentages are illustrated in Figure **4**. The composite materials demonstrated higher microhardness compared to pure PA6. The microhardness of pure PA6 was measured at 10.18 (±0.3) HV, which increased substantially to 11.54 (±0.6) HV with the incorporation of 0.3 wt.%  $Al_2O_3$ . However, as the  $Al_2O_3$  content increased further, the hardness slightly decreased to  $11.36$  ( $\pm 0.9$ ) HV. Despite this slight decrease, the similar standard deviations indicate that the microhardness of both composites remains statistically comparable. The addition of 0.3 wt.% and 1.5 wt.% of  $Al_2O_3$  resulted in hardness improvements of 13.3% and 11.6%, respectively.

 $Al<sub>2</sub>O<sub>3</sub>$  is a ceramic material with a much higher hardness compared to PA6. When these hard particles are embedded in the surface layer of the PA6 matrix, they increase the material's resistance to localized deformation. This leads to a higher microhardness



**Figure 3:** SEM images of 1.5 wt.% of Al2O3/PA6 (**a**) 750 magnifications. EDS mapping of (**b**) aluminum distribution, (**c**) oxygen distribution, and (**d**) nitrogen distribution.



**Figure 4:** Microhardness Vickers value for pristine PA6 and the Al<sub>2</sub>O<sub>3</sub>/PA6 composite.

value because the harder  $Al_2O_3$  particles reduce the indentation depth during testing. The slight reduction in hardness at higher levels of  $Al_2O_3$  reinforcement can be attributed to the uneven distribution of  $Al_2O_3$ particles within the matrix. This uneven dispersion likely leads to localized agglomeration, which weakens the overall structural integrity [15, 16]. The increased

variability in the hardness measurements, as evidenced by the larger standard deviation of ±0.9 compared to ±0.6 at lower reinforcement levels, further supports the observation that poor particle distribution contributes to inconsistent mechanical performance across the composite.



**Figure 5:** Ultimate tensile stress value for pristine PA6 and the Al<sub>2</sub>O<sub>3</sub>/PA6 composite.

### **3.3. Mechanical Properties**

The ultimate tensile strength, as presented in Figure **5**, was evaluated using eight tensile specimens. The base material exhibited a strength of  $52.91 \pm 3.48$ MPa. Upon adding a small quantity of reinforcement, the strength increased to  $56.77 \pm 2.91$  MPa. However, with the introduction of 1.5 wt.% reinforcement into the matrix, a slight decrease was observed, with the tensile strength reducing to  $53.98 \pm 1.04$  MPa. This behavior is consistent with the hardness measurements, where the composite with a low level of  $Al_2O_3$  exhibited the highest performance, and overall, the composite outperformed the pure material. The strength improvement was 7.3% and 2%, adding 0.3 wt.% and 1.5 wt.% of  $Al_2O_3$ , respectively. Nevertheless, agglomeration phenomena also impact the tensile test, like the microhardness. These results further prove that 0.3 wt.% is the optimal or the limit value for the  $Al_2O_3$ .

The strength of a composite is greatly affected by the bonding strength at the interface between the fillers and the matrix materials [19]. Tensile test results show that  $Al_2O_3$  particles are distributed evenly within the matrix when proper mixing procedures are applied before compression molding. The combination of heat and compression promotes strong interfacial bonding between the components. The uniform distribution of fine  $Al_2O_3$  particles within the matrix allows the composite to better withstand stress by dispersing the stress more evenly and minimizing areas of highstress concentration.

# **4. CONCLUSION**

In this study, the composite material of  $Al_2O_3/PA6$ was successfully produced using compression molding. The composite containing 0.3 wt.% Al2O3/PA6 showed a significant improvement in microhardness, increasing by 13.3% from 10.18 HV to 11.54 HV. In terms of tensile strength, the maximum enhancement of 7.3% was achieved by incorporating 0.3 wt.% of reinforcement, increasing the strength from 52.91 MPa to 56.77 MPa. With a higher amount of reinforcement, the improvements in hardness and strength were not optimized due to the agglomeration of  $Al_2O_3$ . In summary, the addition of  $Al_2O_3$  particles significantly improved the PA6 performance.

# **DECLARATION OF INTEREST STATEMENT**

The authors declare that there are no conflicts of interest regarding this study.

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